



<b>AEROSPACE STANDARD</b>	<b>AS1814™</b>	<b>REV. F</b>
	Issued 1983-07 Reaffirmed 2003-04 Revised 2020-07	
Superseding AS1814E		
Terminology for Titanium Microstructures		

## RATIONALE

AS1814F is a Five-Year Review and update of this specification that revises blocky alpha (2.2.18) and the caption for Figure 4.

### 1. SCOPE

- 1.1 This list of terms, with accompanying photomicrographs where appropriate, is intended as a guide for use in the preparation of material specifications.
- 1.2 The terms and photomicrographs are intended to present definitions only; they do not define either acceptance limits or minimum standards of quality.
- 1.3 Listings are not grouped by specific alloys or conditions and represent the typical microstructures wherever they occur.
- 1.4 Etchants used for the microstructures shown are stated. Where "Krolls" is stated, the composition is 10 mL HF, 30 mL HNO<sub>3</sub>, and 50 mL water (H<sub>2</sub>O).
- 1.5 Other common etchants are listed in ASTM E407, Microetching Metals and Alloys.

### 2. REFERENCES

#### 2.1 Applicable Documents

The following publications form a part of this document to the extent specified herein. The latest issue of SAE publications shall apply. The applicable issue of other publications shall be the issue in effect on the date of the purchase order. In the event of conflict between the text of this document and references cited herein, the text of this document takes precedence. Nothing in this document, however, supersedes applicable laws and regulations unless a specific exemption has been obtained.

#### 2.1.1 ASTM Publications

Available from ASTM International, 100 Barr Harbor Drive, P.O. Box C700, West Conshohocken, PA 19428-2959, Tel: 610-832-9585, [www.astm.org](http://www.astm.org).

ASTM E407 Microetching Metals and Alloys

SAE Technical Standards Board Rules provide that: "This report is published by SAE to advance the state of technical and engineering sciences. The use of this report is entirely voluntary, and its applicability and suitability for any particular use, including any patent infringement arising therefrom, is the sole responsibility of the user."

SAE reviews each technical report at least every five years at which time it may be revised, reaffirmed, stabilized, or cancelled. SAE invites your written comments and suggestions.

Copyright © 2020 SAE International

All rights reserved. No part of this publication may be reproduced, stored in a retrieval system or transmitted, in any form or by any means, electronic, mechanical, photocopying, recording, or otherwise, without the prior written permission of SAE.

**TO PLACE A DOCUMENT ORDER:**  
Tel: 877-606-7323 (inside USA and Canada)  
Tel: +1 724-776-4970 (outside USA)  
Fax: 724-776-0790  
Email: [CustomerService@sae.org](mailto:CustomerService@sae.org)  
<http://www.sae.org>

SAE WEB ADDRESS:

For more information on this standard, visit  
<https://www.sae.org/standards/content/AS1814F/>

## 2.2 Terminology

- 2.2.1 ACICULAR ALPHA: A product of nucleation and growth or an athermal (martensitic) transformation from beta to the lower temperature allotropic alpha phase. It may be needle-like, lenticular, or flattened bar morphology in three dimensions. Its typical aspect ratio is about 10:1 (Figure 1).
- 2.2.2 AGED BETA: A beta matrix in which alpha, typically fine, has precipitated as a result of aging or cooling from a temperature high in the alpha-beta phase field (Figure 2).
- 2.2.3 ALPHA: The low temperature allotrope of titanium with a hexagonal, close-packed crystal structure (Figure 3).
- 2.2.4 ALPHA 2 STRUCTURE: A structure consisting of an ordered alpha phase, such as  $Ti_3$  (Al, Sn) found in highly stabilized alpha. Defined by selected area diffraction, not optical metallography.
- 2.2.5 ALPHA-BETA STRUCTURE: A microstructure which contains both alpha and beta as the principal phases at a specific temperature. It is composed of alpha, transformed beta, and retained beta. Structure shown in Figure 4 is typical of mill annealed Ti 6Al-4V; similar structure shown in Figure 18 is more typical of recrystallization annealed Ti 6Al-4V.
- 2.2.6 ALPHA CASE: The oxygen enriched, alpha-stabilized surface which results from elevated temperature exposure to environments containing oxygen or air (Figures 5A, 5B, and 5C). Alpha case is normally hard, brittle, and considered detrimental.
- 2.2.7 ALPHA PRIME: A supersaturated, acicular nonequilibrium hexagonal phase formed by a diffusionless transformation of the beta phase. It occurs when cooling rates are too high to permit transformation by nucleation and growth. It exhibits an aspect ratio of 10:1 or greater. Also known as martensite or martensite alpha (Figure 6).
- 2.2.8 ALPHA DOUBLE PRIME (ORTHORHOMBIC MARTENSITE): A supersaturated nonequilibrium orthorhombic phase formed by a diffusionless transformation of the beta phase in certain alloys. It occurs when cooling rates are too high to permit transformation by nucleation and growth. It may be strain induced during working operations and may be avoided by appropriate in-process annealing treatments.
- 2.2.9 ALPHA STABILIZER: An alloying element which dissolves preferentially in the alpha phase and raises the alpha-beta transformation temperature. Aluminum is the most commonly used alpha stabilizer. Interstitial elements such as oxygen and nitrogen are also potent alpha stabilizing elements.
- 2.2.10 ALPHA STRINGER: Platelet alpha that has been elongated and distorted by metal working but not broken up or recrystallized. Also called "Wormy Alpha" or "Stringy Alpha" (Figure 19).
- 2.2.11 ALPHA-TRANSUS: The temperature that alpha begins to revert to beta.
- 2.2.12 BASKETWEAVE: Alpha platelets, with or without interweaved beta platelets, that occur in colonies. Also known as Widmanstätten. Forms during cooling through the beta transus at intermediate cooling rates (Figure 7A).
- 2.2.13 BETA: The allotrope of titanium with a body-centered cubic crystal structure occurring at temperatures between the solidification of molten titanium and the beta transus, i.e., the high temperature allotrope.
- 2.2.14 BETA EUTECTOID STABILIZER: An alloying element that dissolves preferentially in the beta phase, lowers the alpha-beta to beta transformation temperature, under equilibrium conditions, and results in the beta decomposition to alpha plus a compound. This is a eutectoid reaction. Commonly used beta eutectoid forming elements are iron, nickel, chromium, and manganese.
- 2.2.15 BETA FLECK: Beta flecks have reduced amounts of, or no, primary alpha which may exhibit a morphology different from the primary alpha in the surrounding alpha-beta matrix and/or absence of alpha stabilizers such as oxygen or aluminum. The flecks are then seen by the reduced or lack of alpha within the beta fleck (Figure 8).

- 2.2.16 **BETA ISOMORPHOUS STABILIZER:** An alloying element that is soluble in beta titanium in all proportions. It lowers the alpha-beta to beta transformation temperature without a eutectoid reaction and forms a continuous series of solid solutions with beta titanium. Commonly used beta isomorphous forming elements are vanadium, molybdenum, and zirconium.
- 2.2.17 **BETA TRANSUS:** The temperature that designates the phase boundary between the alpha plus beta and beta fields. Commercially pure grades transform in a range of 1630 to 1760 °F (890 to 960 °C) depending upon oxygen and iron content. In general, aircraft alloys vary in transformation temperature from 1380 to 1900 °F (750 to 1040 °C).
- 2.2.18 **BLOCKY ALPHA:** Alpha phase which is considerably larger and more polygonal in appearance than the primary alpha present. It may result from extended exposure high in the alpha-beta phase field following slow cooling through the beta transus during forging or heat treating operations. It may be removed by beta recrystallization or by all-beta working followed by further alpha-beta work. May accompany grain boundary alpha or even have its origin as large grain boundary alpha or coarse alpha platelets. Microhardness not significantly different from surrounding normal alpha-beta matrix (Figure 9).
- 2.2.19 **COLONIES:** Regions within prior beta grains with alpha platelets having nearly identical orientations. In commercially pure titanium, colonies often have serrated boundaries. Colonies arise as transformation products during cooling from the beta field at cooling rates slow enough to allow platelet nucleation and growth (Figures 7A and 7B).
- 2.2.20 **ELONGATED ALPHA:** The hexagonal crystal phase appearing as stringer-like arrays, considerably larger in appearance than the primary alpha. Commonly exhibits an aspect ratio of 3:1 or higher (Figures 9, 10, and 11).
- 2.2.21 **EQUIAXED STRUCTURE:** A polygonal or spheroidal microstructural feature having approximately equal dimensions in all directions. In alpha-beta titanium alloys, such a term commonly refers to a microstructure in which most of the alpha phase appears spheroidal, primarily in the transverse direction (Figures 3, 4, and 18).
- 2.2.22 **FREQUENCY OF OCCURRENCE:** A referee determination by viewing 50 fields, 4 x 5 inches (102 x 127 mm), projected at 100X. The number of fields containing the feature of interest is divided by the total number of fields viewed to represent the lot, thus arriving at a percentage.
- 2.2.23 **GAMMA STRUCTURE:** An ordered structure of titanium-aluminum compound with a stoichiometric ratio TiAl and face-centered tetragonal crystal structure.
- 2.2.24 **GLOBULAR ALPHA:** A spheroidal form of equiaxed alpha (Figure 4).
- 2.2.25 **GRAIN BOUNDARY ALPHA:** Alpha outlining prior beta grain boundaries. It may be continuous unless broken up by subsequent work. Also may accompany blocky alpha. Occurs by slow cooling from the beta field into alpha-beta field (Figures 1, 7B, 11, and 12).
- 2.2.26 **HIGH ALUMINUM DEFECT (HAD):** An aluminum-rich alpha stabilized region containing an abnormally large amount of aluminum which may extend across a large number of beta grains. It contains an inordinate fraction of primary alpha but has a microhardness only slightly higher than the adjacent matrix. These are also known as Type II defects (Figures 13 and 14).
- 2.2.27 **HIGH DENSITY INCLUSION (HDI):** A region with a concentration of elements, usually tungsten, molybdenum, tungsten carbide, or residuals of high melting point master alloys containing molybdenum, having a higher density than the matrix. Regions are readily detectable by X-ray and will appear brighter than the matrix.
- 2.2.28 **HIGH INTERSTITIAL DEFECT (HID):** Interstitially stabilized alpha phase region of substantially higher hardness and lower ductility than surrounding material. It arises from melting titanium in the presence of nitrogen, oxygen, or carbon. HID typically have a diffusion zone. They are commonly called Type I defects or low-density inclusions (LDI). HIDs often associated with voids and cracks (Figures 15 and 16).
- 2.2.29 **HYDRIDE PHASE:** The phase TiH<sub>x</sub> formed in titanium when the hydrogen content exceeds the solubility limit. Hydrogen and, therefore, hydrides tend to accumulate at areas of high residual tensile stresses (Figures 17A and 17B).

- 2.2.30 INTERSTITIAL ELEMENT: An element with relatively small atomic diameter that can assume position in the interstices of the titanium crystal lattice. Common examples are oxygen, nitrogen, hydrogen, and carbon.
- 2.2.31 INTERGRANULAR BETA: Beta phase situated between alpha grains. It may be at grain corners as in the case of equiaxed alpha type of microstructures in alloys having low beta stabilizer content. Because it is the original phase from which alpha is formed, it is usually the continuous phase in two phase microstructures (Figure 18).
- 2.2.32 INTERMETALLIC COMPOUND: A phase in an alloy system which usually occurs at a definite atomic ratio and exhibits a narrow solubility range, such as  $Ti_2Fe$ ,  $Ti-Ni$ , or  $\alpha_2$ . Nearly all such phases are brittle.
- 2.2.33 MARTENSITE: See "Alpha Prime", "Alpha Double Prime" (orthorhombic martensite in some significantly beta stabilized alloys).
- 2.2.34 MATRIX: The constituent which forms the continuous phase of a two or more phase microstructure.
- 2.2.35 METASTABLE BETA: A nonequilibrium phase composition that can be partially transformed to martensite, alpha, or eutectoid decomposition products with thermal or strain energy activation during subsequent processing or service exposure.
- 2.2.36  $M_f$ : The temperature at which the martensite reaction is complete.
- 2.2.37  $M_s$ : The maximum temperature at which a martensite reaction begins upon cooling from the beta phase or high in the alpha-beta field.
- 2.2.38 OMEGA: A nonequilibrium, submicroscopic phase which can be formed either athermally or isothermally preceding the formation of alpha from beta. It occurs in metastable beta alloys, alpha-plus-beta alloys rich in beta content, and CP titanium, and leads to severe embrittlement in metastable beta alloys or alpha-beta alloys rich in beta content. Athermal omega is believed to form without change in composition and is analogous to martensite. Isothermal omega is generally formed by aging a retained beta structure in the 392 to 932 °F (200 to 500 °C) temperature range.
- 2.2.39 ORDERED STRUCTURE: The orderly or periodic arrangement of solute atoms on the lattice sites of the solvent.
- 2.2.40 PLATELET ALPHA: A relatively coarse acicular alpha, usually with low aspect ratios. This microstructure arises from cooling alpha or alpha-beta alloys at a slow rate from temperatures that a significant fraction of beta phase exists (Figure 10).
- 2.2.41 PRIMARY ALPHA: The allotrope of titanium with a hexagonal, close-packed crystal structure which is retained from the last high temperature alpha-beta heating (Figure 4).
- 2.2.42 PRIOR BETA GRAIN SIZE: Size of beta grains established during the most recent beta field excursion. The grains may be distorted by subsequent subtransus deformation. The beta grain boundaries may be obscured by a superimposed alpha-beta microstructure and detectable only by special techniques (Figure 12).
- 2.2.43 STRINGY ALPHA: See "Alpha Stringer" (Figure 19).
- 2.2.44 SUBSTITUTIONAL ELEMENT: An alloying element with an atom size and other features similar to the titanium atom, which can replace or substitute for the titanium atoms in the lattice and form a significant region of solid solution in the phase diagram. Such elements used in alloying titanium include but are not limited to aluminum, vanadium, molybdenum, chromium, iron, tin, and zirconium.
- 2.2.45 TRANSFORMED BETA: A local or continuous structure comprised of decomposition products arising either by martensitic or by nucleation and growth processes during cooling from either above the beta transus or some temperature high in the alpha-beta phase field. The structure typically consists of alpha platelets which may or may not be separated by beta phase.

2.2.46 TWIN: Two portions of a crystal having a definite crystallographic relationship; one may be regarded as the parent, the other as the twin. The orientation of the twin is either a mirror image of the orientation of the parent about a "twinning plane" or an orientation that can be derived by rotating the twin portion about a "twinning axis" (Figure 3).

2.2.47 WIDMANSTATTEN STRUCTURE: See "Basketweave."

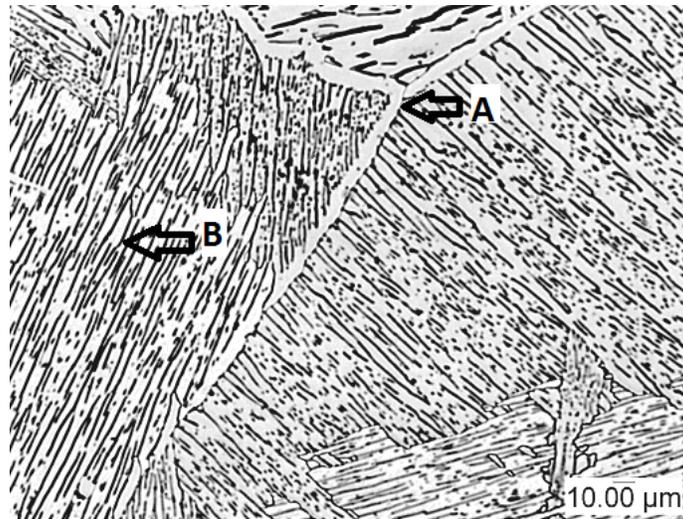
2.2.48 WORMY ALPHA: See "Alpha Stringer" (Figure 19).

### 3. NOTES

#### 3.1 Revision Indicator

A change bar (|) located in the left margin is for the convenience of the user in locating areas where technical revisions, not editorial changes, have been made to the previous issue of this document. An (R) symbol to the left of the document title indicates a complete revision of the document, including technical revisions. Change bars and (R) are not used in original publications, nor in documents that contain editorial changes only..

SAENORM.COM : Click to view the full PDF of as1814f



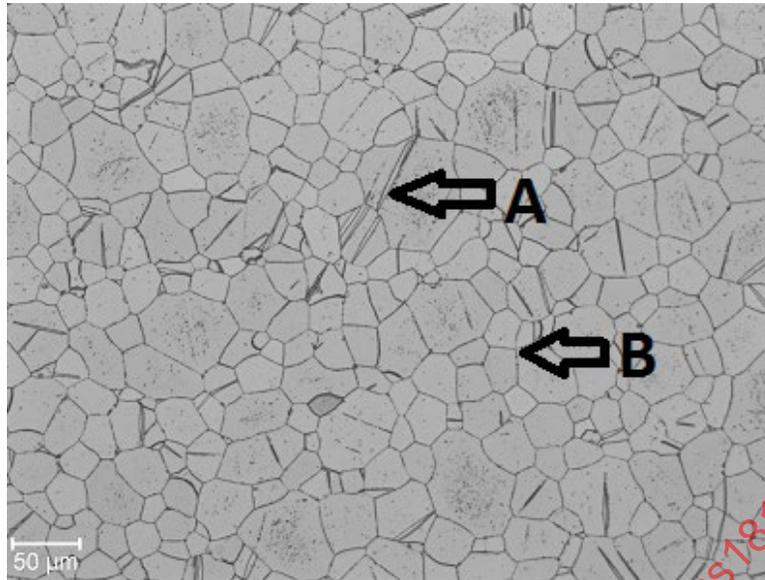
ETCHANT: KROLL'S 300X  
 ALLOY 6Al-4V. ACICULAR ALPHA COLONIES IN PRIOR BETA GRAINS.  
 "A" = GRAIN BOUNDARY ALPHA, "B" = ACICULAR ALPHA COLONIES.

**Figure 1**



ETCHANT: KROLL'S 500X  
 ALLOY 3Al-8V-6Cr-4Mo-4Zr. AGED BETA. BETA MATRIX WITH ALPHA PRECIPITATES.  
 SOLUTION ANNEALED AT 1450 °F (788 °C), AIR COOLED, AGED AT 950 °F (510 °C), 24 HOURS.

**Figure 2**



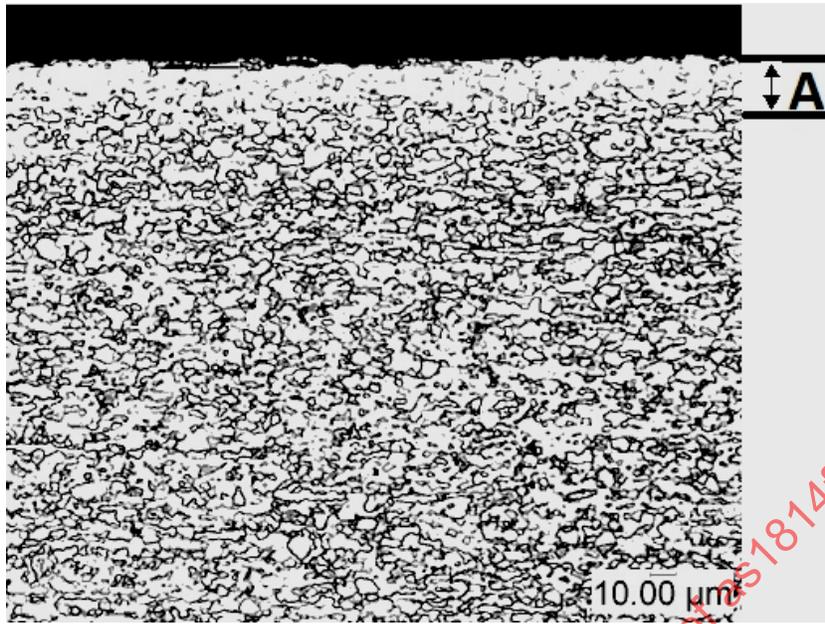
ETCHANT: 2%HF 200X  
 ALLOY COMMERCIALY PURE (CP). TWIN "A" IN EQUIAXED ALPHA "B"

**Figure 3**



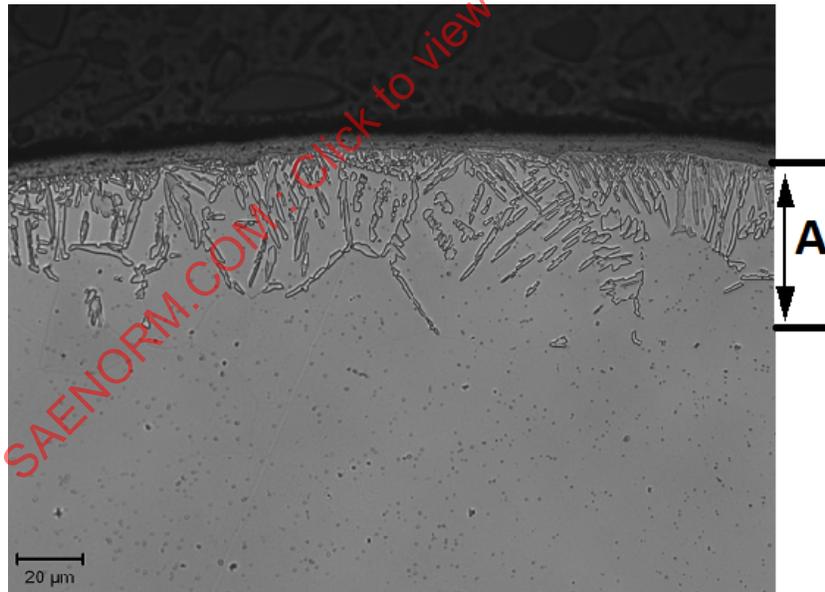
ETCHANT: KROLL'S 600X  
 ALLOY 6Al-4V. PRIMARY ALPHA (LIGHT ETCHING) IN A TRANSFORMED BETA MATRIX (DARK ETCHING)  
 CONTAINING FINE ACICULAR ALPHA

**Figure 4**



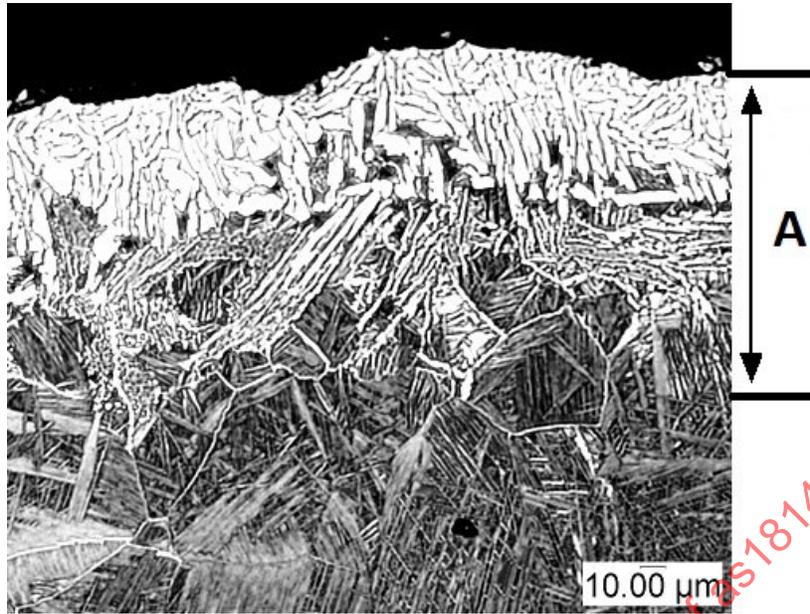
ETCHANT: 2% HF IN SATURATED SOLUTION OF OXALIC ACID 300X  
ALLOY 6Al-4V. ALPHA CASE "A" ON ALPHA BETA ALLOY.

*Figure 5A*



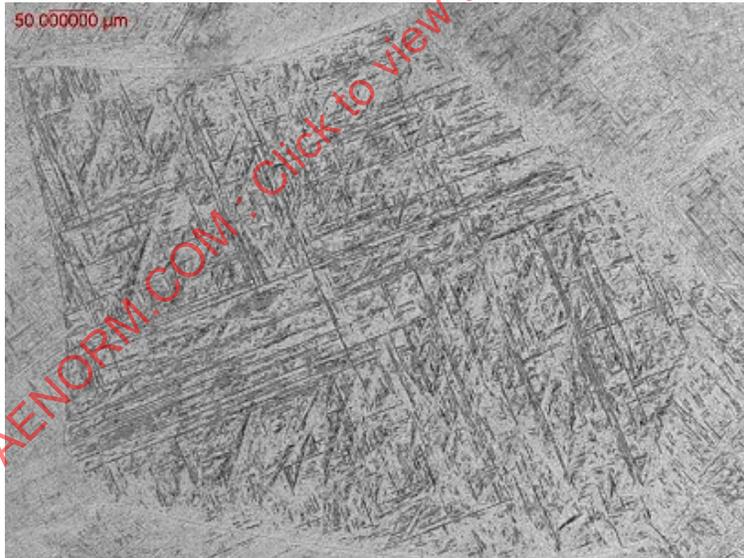
ETCHANT: KROLL'S 500X  
ALLOY 3Al-8V-6Cr-4Zr-4Mo. ALPHA CASE "A" ON BETA ALLOY.

*Figure 5B*



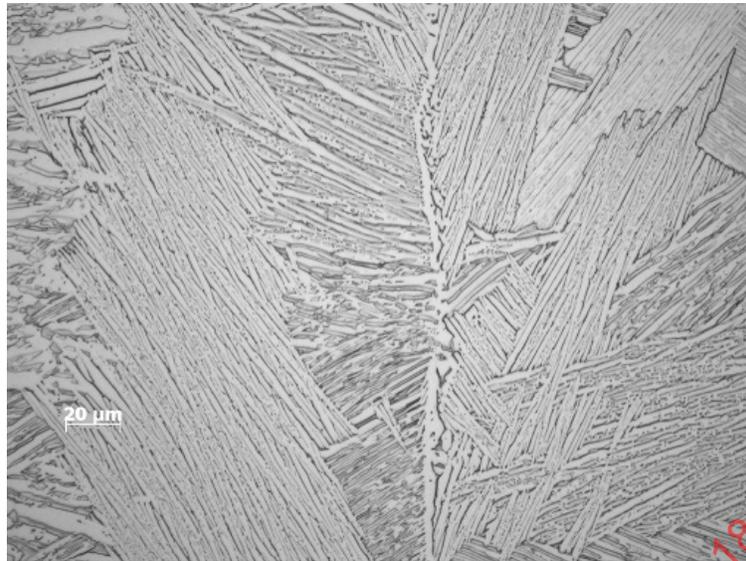
ETCHANT: KROLL'S 300X  
ALLOY 6Al-2Sn-4Zr-6Mo. ALPHA CASE "A" ON ALPHA BETA ALLOY.

**Figure 5C**



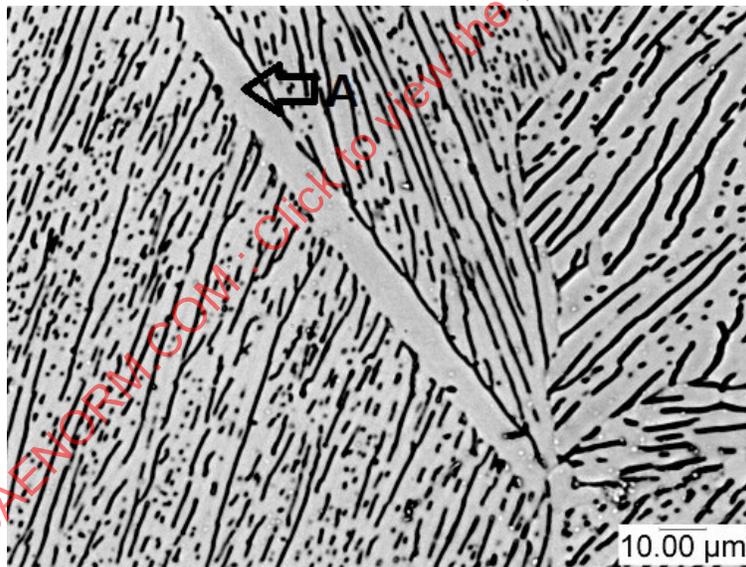
ETCHANT: KROLLS 100X  
ALLOY 6Al-4V. ALPHA PRIME (MARTENSITE).

**Figure 6**



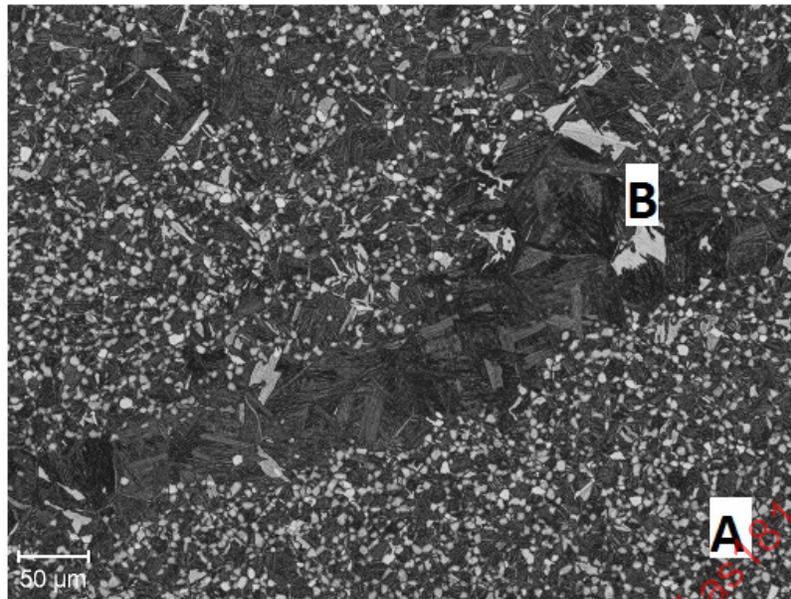
ETCHANT: KROLL'S 100X  
ALLOY 6Al-4V. COLONIES OF ALPHA IN BASKETWEAVE STRUCTURE IN BAR.

*Figure 7A*



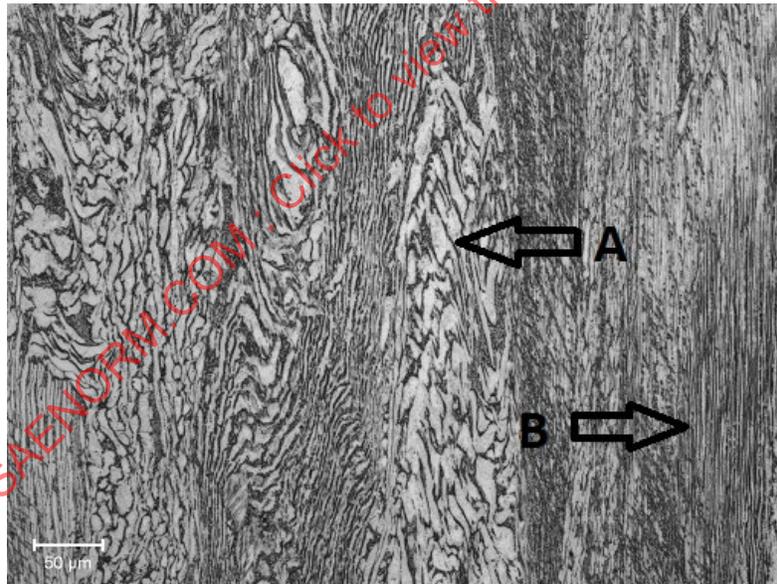
ETCHANT: KROLL'S 600X  
ALLOY 6Al-4V. COLONIES OF ALPHA IN BASKETWEAVE STRUCTURE WITH "A" GRAIN BOUNDARY ALPHA.

*Figure 7B*



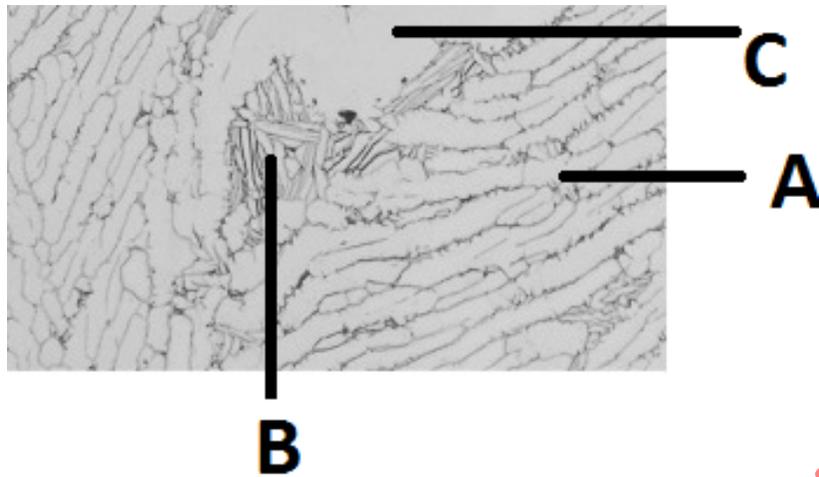
ETCHANT: AMMONIUM BIFLOURIDE 200X  
ALLOY 6Al-4V BETA FLECK "B" in ALPHA BETA MATRIX "A"

**Figure 8**



ETCHANT: KROLL'S 200X  
ALLOY 6Al-4V. BLOCKY ALPHA "A" WITH ELONGATED ALPHA "B".

**Figure 9**



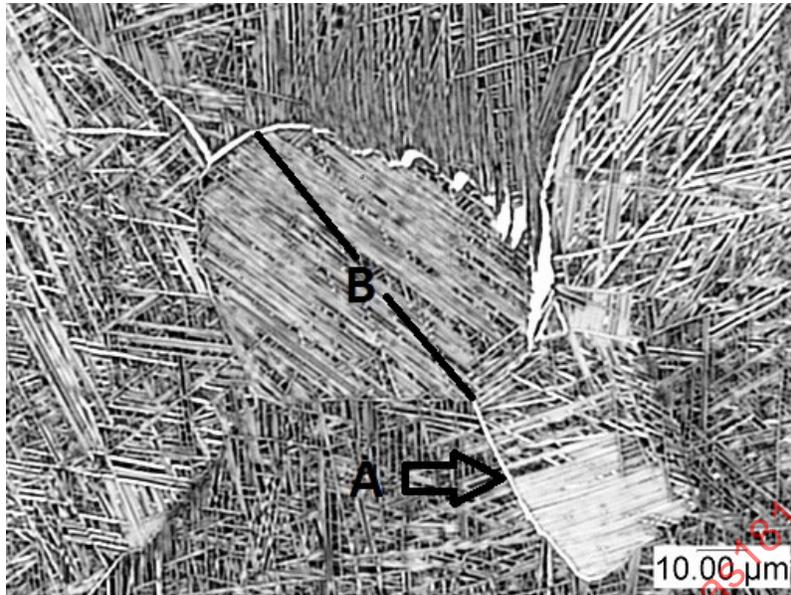
ETCHANT: KROLL'S 200X  
 ALLOY 6Al-4V. ELONGATED ALPHA 'A' AND ACICULAR ALPHA 'B'.  
 BLOCKY ALPHA 'C' IS SEEN IN THE CENTRAL AREA (ALSO KNOWN AS GLOBULAR ALPHA).

*Figure 10*



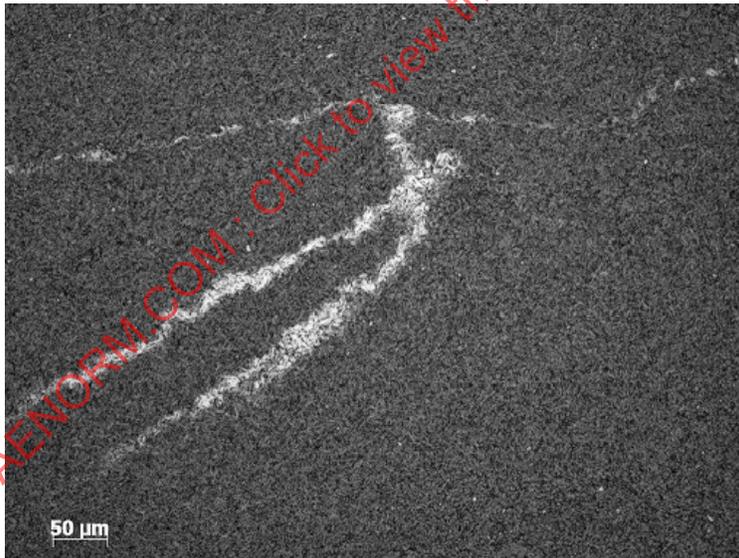
ETCHANT: KROLL'S 100X  
 ALLOY 6Al-2Sn-4Zr-6Mo. GRAIN BOUNDARY ALPHA "A" WITH ELONGATED  
 ALPHA "B".

*Figure 11*



ETCHANT: KROLL'S 600X  
ALLOY 6Al-2Sn-4Zr-6Mo. GRAIN BOUNDARY ALPHA "A" SURROUNDING PRIOR BETA GRAINS "B".

*Figure 12*



ETCHANT: 2% HF + AMMONIUM BIFLOURIDE 200X  
ALLOY 6Al-4V. ALPHA SEGREGATION / HIGH ALUMINUM DEFECT (HAD).

*Figure 13*